



Core level spectroscopy: XPS, XAS, EELS, XES (XSPEC, TELNES)

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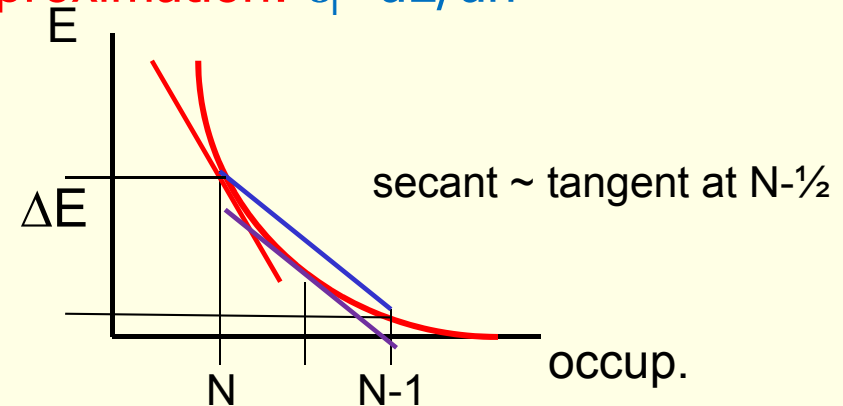
TU Vienna



XPS, core-level shifts



- Ionization potential of core- e^- , $IP = E^{\text{tot}}(N) - E^{\text{tot}}(N-1)$
 - *gives information on charge state of the atom*
- core-eigenvalues ε_i are NOT a good approximation: $\varepsilon_i = dE/dn$
- Slater's "transition state":
 - *core-eigenvalues ε_i for half occupancy*



- Δ -SCF-calculation with and without core-hole: $E^{\text{tot}}(N) - E^{\text{tot}}(N-1)$
 - *supercells to reduce hole-hole interaction*

C,N 1s	exp.(eV)	ε_i	Δ -SCF
TiC	281.5	264.7	281.9
Ti ₄ C ₄	281.5	263.3	281.1
TiN	397.0	377.5	397.1

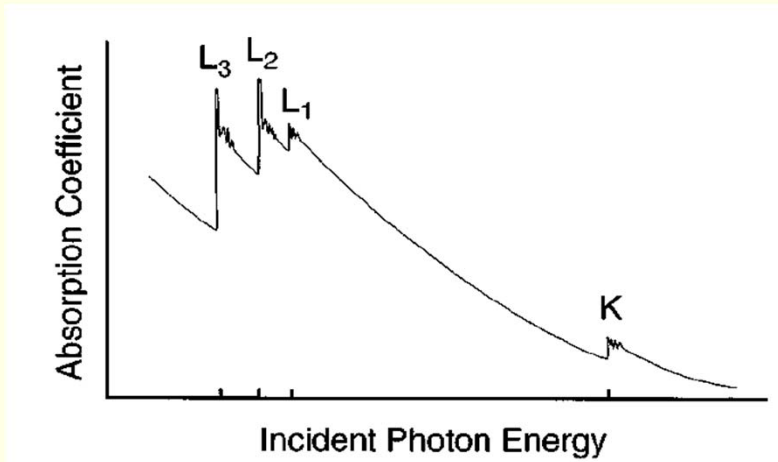


XAS (XANES), EELS (ELNES), XES:



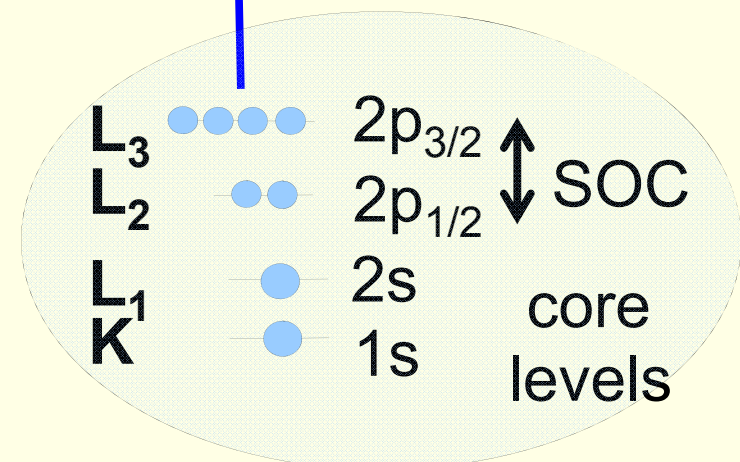
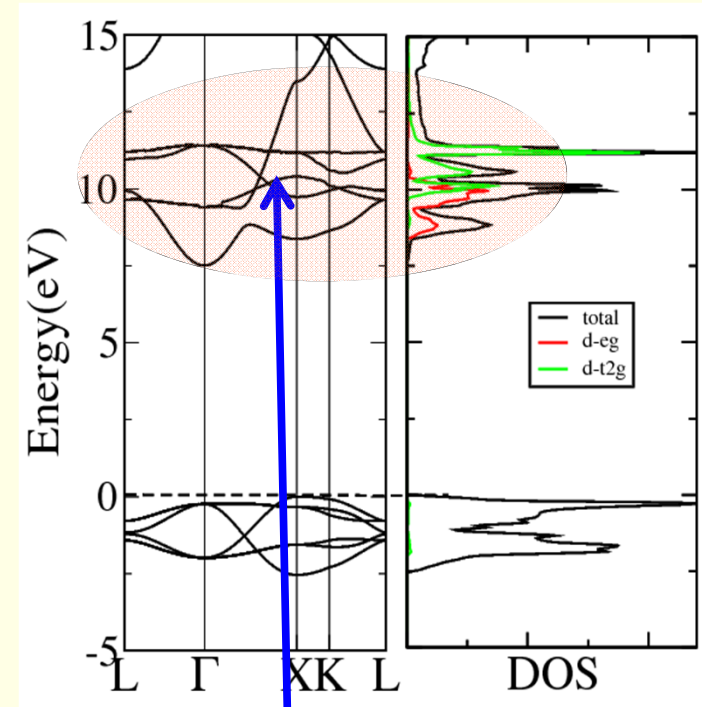
- core electrons are excited into a conduction band
- Each core shell introduces an absorption edge, (they are indexed by the principal number of a core level)

$K-1s$, L_1-2s , $L_2-2p_{1/2}$, $L_3-p_{3/2}$



- **XES:**

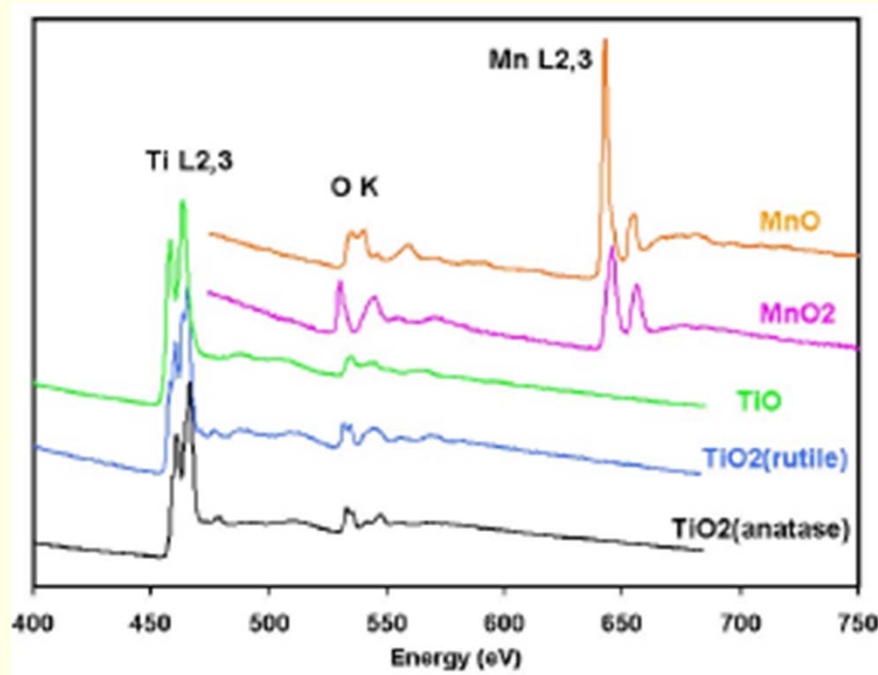
knock out a core electron, **valence electron** fills core hole and **$h\nu$** is emitted





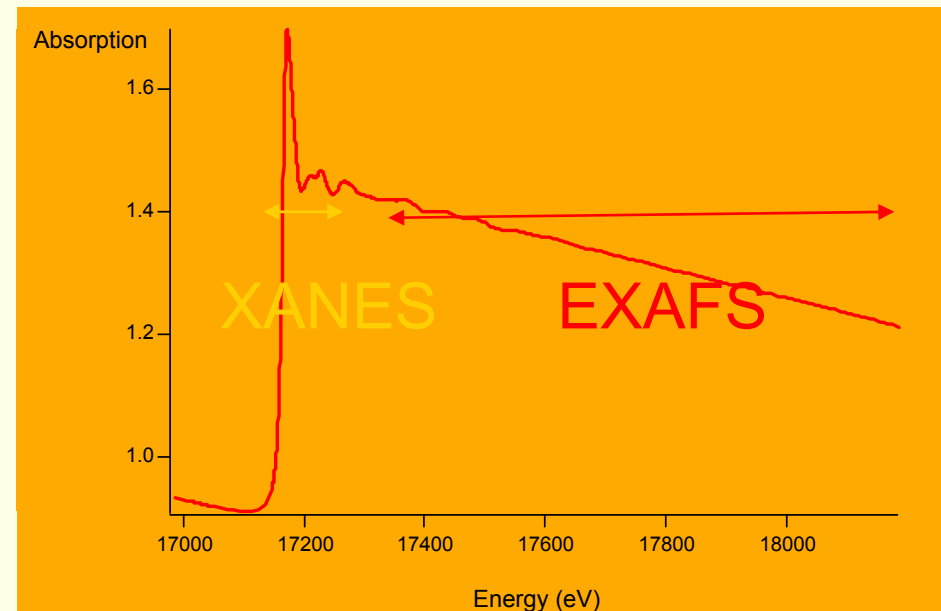
X-ray absorption spectroscopy - XAS

Electron energy loss spectroscopy - EELS



EELS spectrum of various TM oxides

X-ray Absorption
Near-Edge Structure
Extended X-ray
Absorption Fine Structure





Difference between EELS and XAS



XAS: synchrotron



EELS: microscope



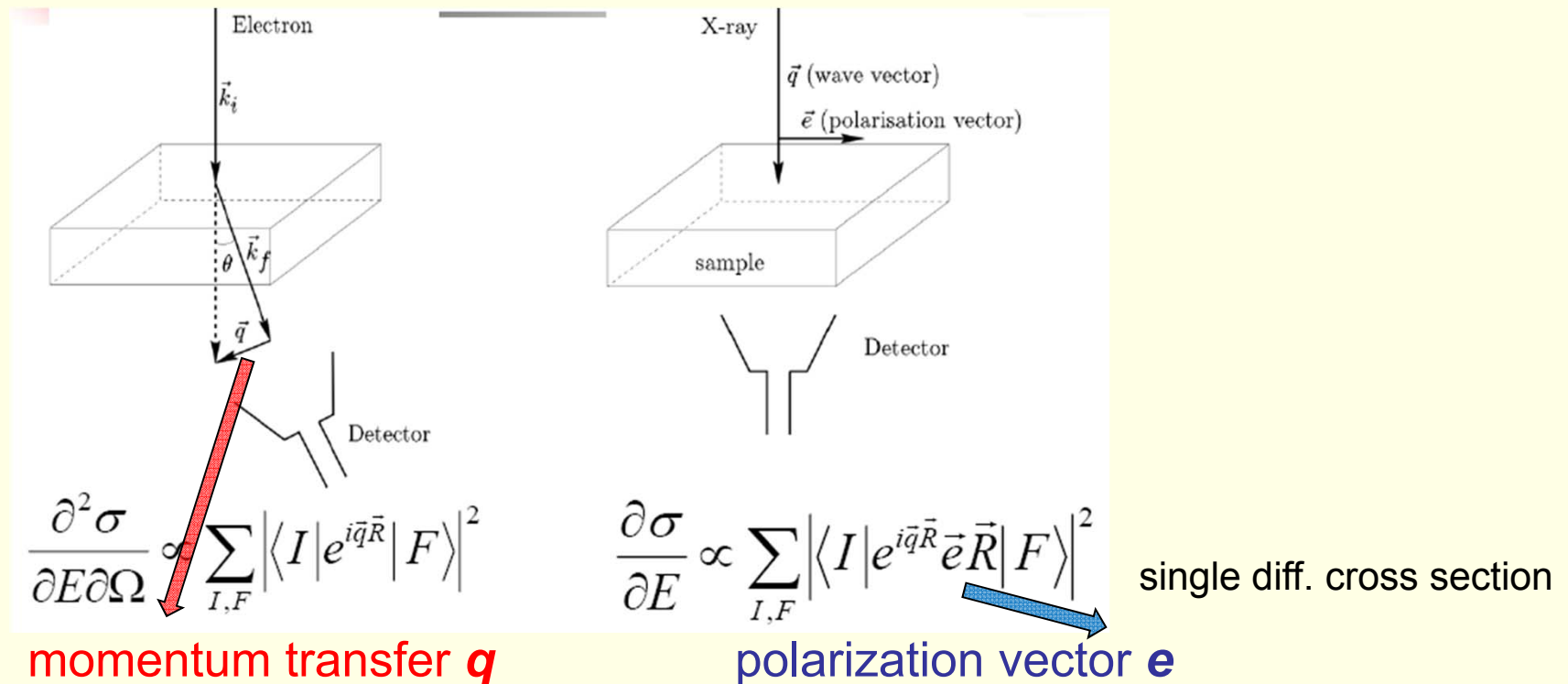


XAS vs. EELS: theory



- transition described by Fermis "golden rule" between initial (core) and final (conduction-band) state and the e⁻ or photon
- double differential cross section:

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega}(E, \mathbf{Q}) = \zeta \sum_{I, F} \frac{k_F}{k_I} \left| \langle I k_I | V | k_F F \rangle \right|^2 \delta(E_I - E_F) \quad \text{E - conservation}$$





dipole approximation



$$\vec{q}\vec{R} \ll 1 \rightarrow e^{i\vec{q}\vec{R}} = 1 + i\vec{q}\vec{R} + \frac{(\vec{q}\vec{R})^2}{2!} + \dots$$

EELS

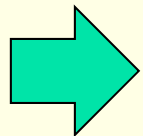
$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} \propto \sum_{I,F} \left| \langle I | \vec{q}\vec{R} | F \rangle \right|^2$$

XAS

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} \propto \sum_{I,F} \left| \langle I | \vec{\epsilon}\vec{R} | F \rangle \right|^2$$

The **polarization vector** in XAS plays the same role as **momentum transfer** in (nonrelativistic) ELNES within the dipole approximation.

(TELNES3 can also handle non-dipole transitions + relativistic corrections)



core-valence spectroscopies give information on the **local DOS** (because of $\langle \Psi_{\text{core}} | r | \Psi_{\text{val}} \rangle$) of angular momentum character $l \pm 1$



“Final state rule”:



“Final state” determines the spectrum:

- **Emission spectroscopy:**

Final state has filled core, but valence hole.

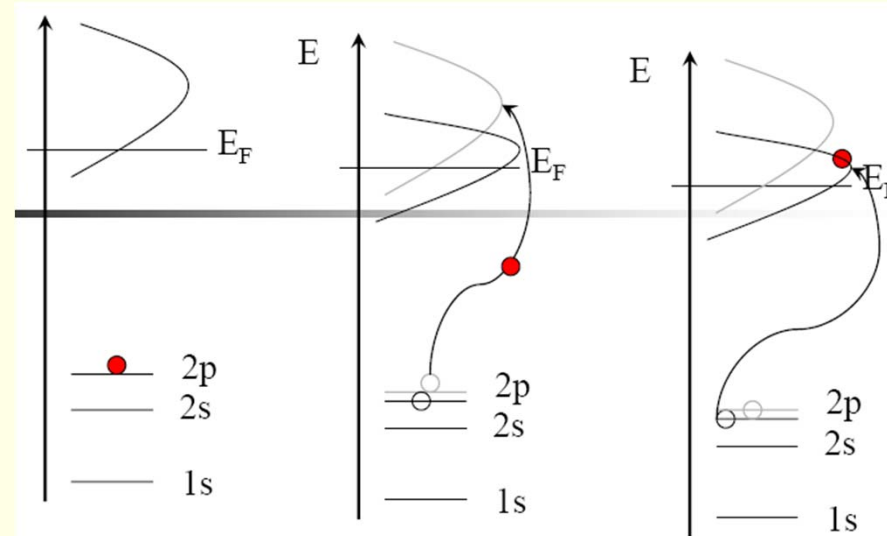
This is usually well screened, thus one “sees” the **groundstate**.

- **Absorption spectroscopy:**

Final state has a “hole” in core state, but additional e^- in conduction band.

Core-hole has large effect on the spectrum

- **electron – hole interaction, “excitonic effects”**





Treating the core hole within WIEN2k



- No core hole (= ground state, sudden approximation)
 - *usually not a good approximation (maybe in metals ?)*
- Z+1 approximation (eg., replace C by N)
 - *also not very good*
- **Core-hole (supercell) calculations:**
 - *Remove 1 core electron on ONE atom in the supercell, add 1 electron to conduction band*
 - *Remove 1 core electron, add 1 electron as uniform background charge*
 - **considers statically screened e^- - h coulomb correlation**
 - **Fractional core hole** (consider different screening)
- Explicit treatment of electron-hole interaction (excitonic effects) using Bethe-Salpeter equation (BSE)



"Final state rule" + core hole:

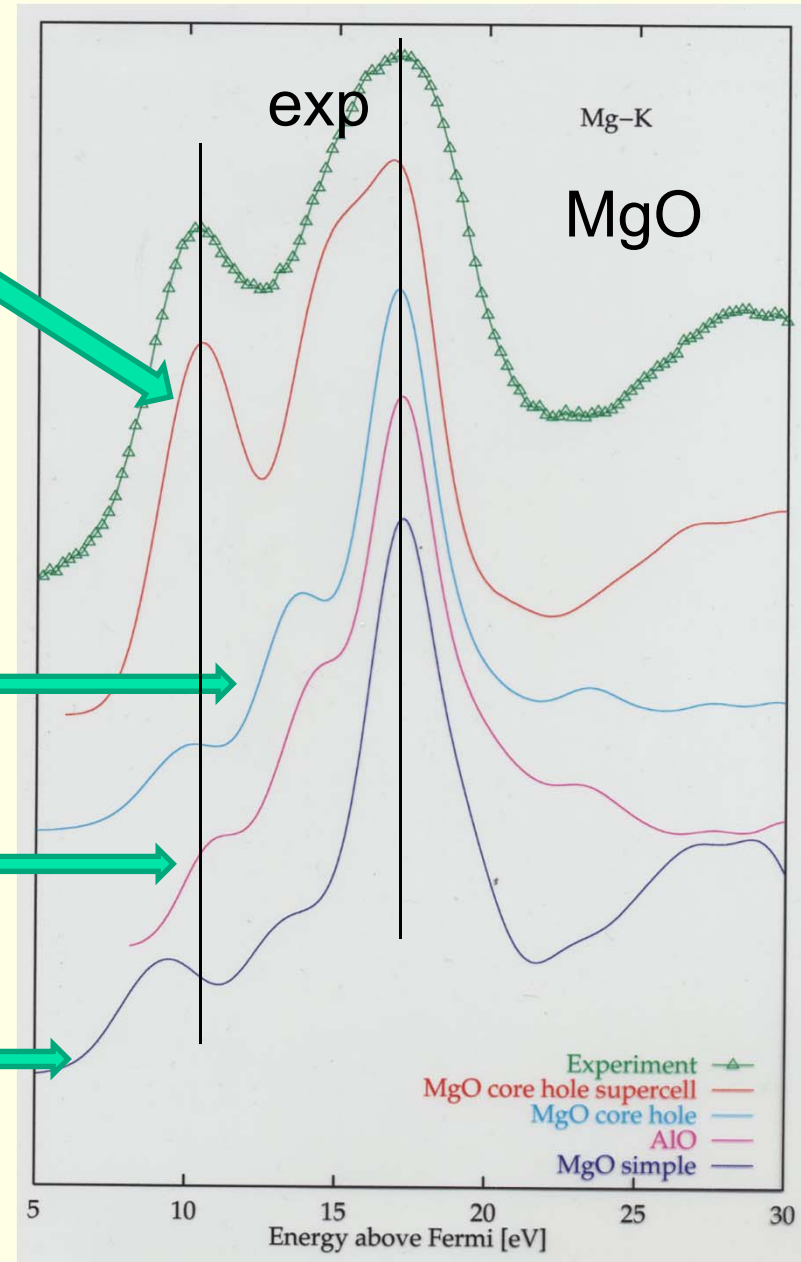


→ 2x2x2 **supercell calculation**, with core hole in **one** of the Mg atoms. This allows the conduction state to relax (adjust to the larger **effective** nuclear charge), but also to have static screening from the environment.

core hole, no supercell:

Z+1 (AlO)

groundstate





XAS / TELNES in WIEN2k



define your structure (structgen)

initialize calculation (init_lapw)

run scf-cycle (run_lapw)

geometry optimization of your structure (min_lapw)

generate supercell (x supercell)

initialize supercell structure, define core hole/add valence e⁻

run scf-cycle

remove extra valence e⁻

run XSPEC / TELNES3 task

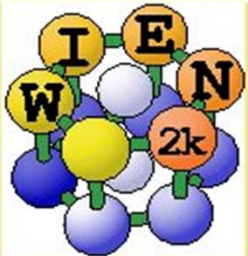


XSPEC-task



Session: [magnetite](#)
/area51/pblaha/lapw/correlated/magnetite

16:42:50 ide
[\[refresh.\]](#) | [\[no refresh.\]](#)



[Execution >>](#)

- [\[StructGen™ \]](#)
- [\[view structure \]](#)
- [\[initialize calc. \]](#)
- [\[run SCF \]](#)
- [\[single prog. \]](#)
- [\[optimize\(V,c/a\) \]](#)
- [\[mini. positions \]](#)

[\[Utils. >> \]](#)

[\[<< Tasks \]](#)

- [\[El. Dens. \]](#)
- [\[DCS \]](#)
- [\[XSPEC \]](#)
- [\[TELNES.2 \]](#)
- [\[OPTIC \]](#)
- [\[Bandstructure \]](#)

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- [\[struct file\(s\) \]](#)
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- [\[output files \]](#)
- [\[SCF files \]](#)

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- [\[change info \]](#)

[\[Configuration \]](#)

Usersguide

- [\[html-Version \]](#)
- [\[pdf-Version \]](#)

XSPEC

[\[Spin UP \]](#) | [\[Spin DOWN \]](#)

Spin UP selected.

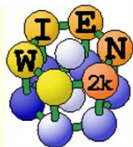
If you want to include states with higher energy

- [Edit in1](#)
- Calculate eigenvalues interactively
- Calculate eigenvalues interactively
- Calculate partial charges interactively
- [Edit input-file for XSPEC](#)
- Calculate X-ray spectra interactively
- [Plot XSPEC](#)

```

Title: Atom 1 L3 absorption spectrum
1          (atom)
2          (n core)
1          (l core)
0,0.5,0.5  (split, Int1, Int2)
-2,0.02,15 (EMIN,DE,EMAX)
ABS        (type of spectrum)
1.00       (S)
2.0        (gamma0)
1.50       (W only for EMIS)
AUTO       (AUTO or MANually select Energy)
-6.93
-10.16
-13.9

```

TELNES3 task



Session: [\[rutile \]](#)
/psi11/pblaha/lapw/bulk/Rutil_exp

TELNES3

[edit Rutil_exp.innes](#) Edit input

Only if you want to include

[edit Rutil_exp.in1](#) Edit in 1

[x lapw1](#) Calculate eigenvalues

[x qtl -telnes](#) Calculate partial charges

[x telnes3](#) Calculate ELNES

[view Rutil_exp.outputtelnes](#) display

[edit Rutil_exp.inb](#) Edit input file

[x broadening](#) Broaden the spectrum

[plot](#) Plot ELNES

[save_eels](#) Save an elnes case

Execution >>

[\[StructGen™ \]](#)
[\[view structure \]](#)
[\[initialize calc. \]](#)
[\[run SCF \]](#)
[\[single prog. \]](#)
[\[optimize\(V,c/a\) \]](#)
[\[mini. positions \]](#)

[Utils. >>]

[\[<< Tasks \]](#)
[\[El. Dens. \]](#)
[\[DOS \]](#)
[\[XSPEC \]](#)
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InnesGen™ for TELNES3

Title:

Atom:

Edge: (n= l=)

Edge onset: eV

Beam energy: keV

Energy grid: eV to eV in steps of eV

Collection s.a.: mrad

Convergence s.a.: mrad

Spectrometer broadening eV Q-mesh: NR= NT=

Advanced settings:

Branching ratio: (statistical if empty)

Spinorbit splitting of core state (eV): (calculated if empty)

Orientation sensitive: $\alpha =$ °, $\beta =$ °, $\gamma =$ °

Integrate over equivalent atoms: to (all eq. atoms if empty)

Detector position: θ_x mrad, θ_y mrad

Modus:

Initialization: Calculate DOS

write DOS

Calculate rotation matrices

write rotation matrices

Verbosity:

File headers:

Interaction potential:

Q-grid: U $\theta_0 =$ (not used for uniform grid)

Interaction order:

Final state selection rule:

Extend potential beyond Rmt: rmax= bohr

Set Fermi energy manually: EF= Ry

Read core state wavefunction: filename=

Read final state wavefunctions: filename=

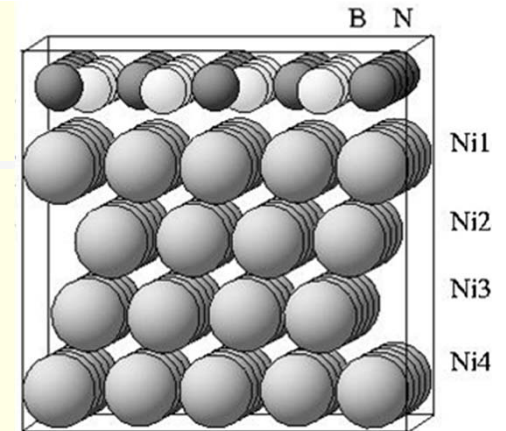
Calculate DOS only

NBTOT:

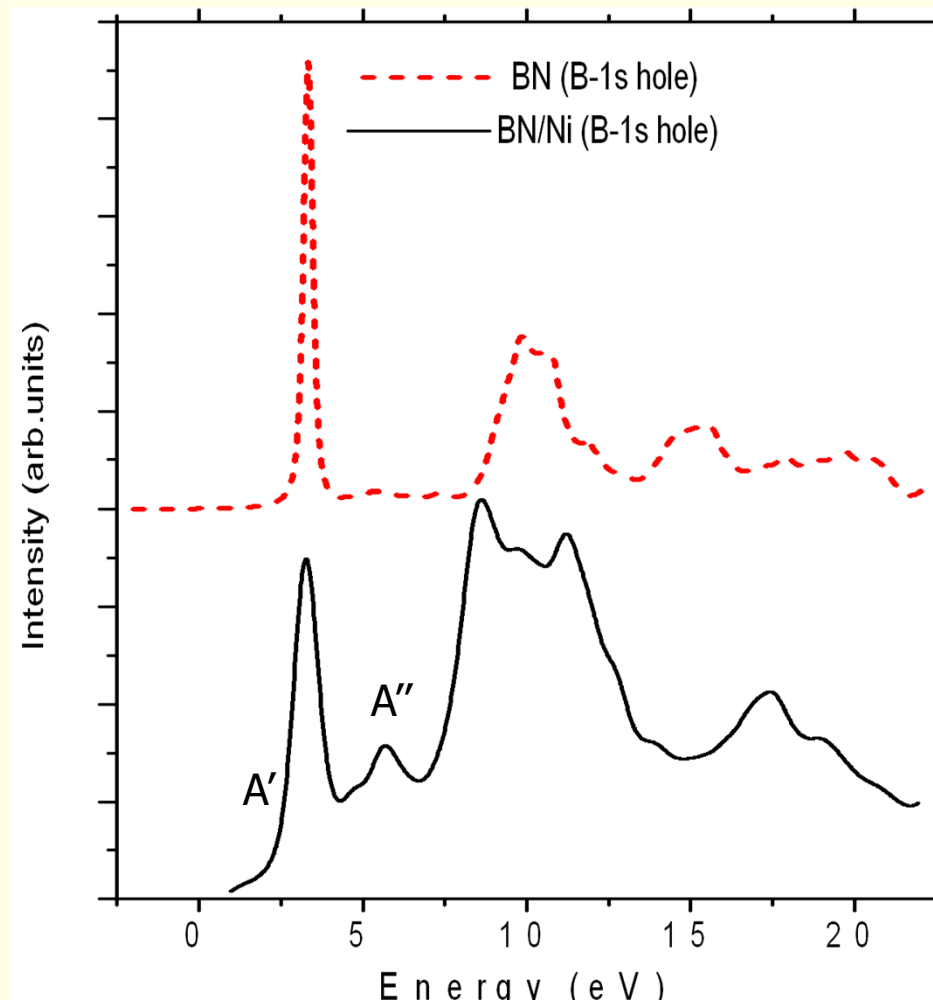
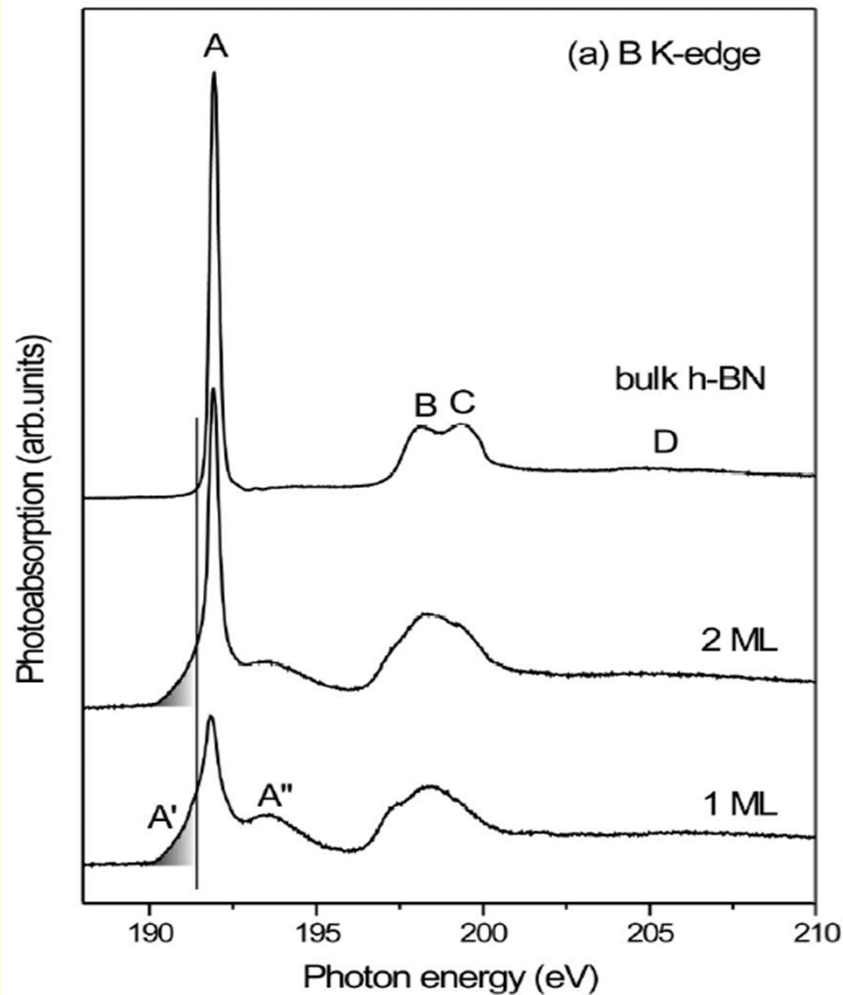
[save](#)



B-K XANES in h-BN/Ni(111)

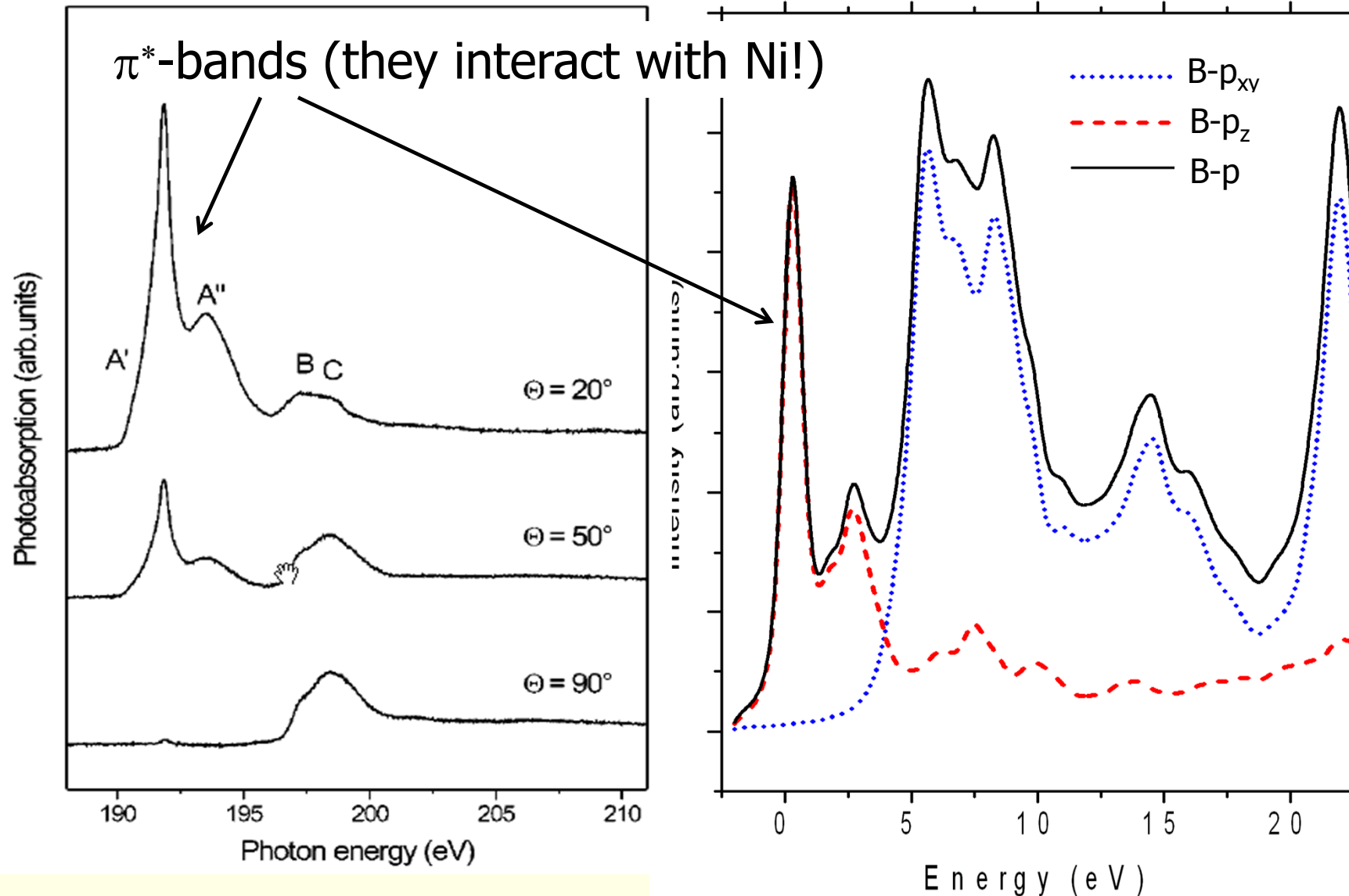


- **B-K edge in BN and BN/Ni(111)**
- Preobrajenski et al, PRB70, 165404 (2004): "The experiments contradict recent DFT calculations by Grad et al."





Angle dependency of B-K edge in h-BN/Ni(111)

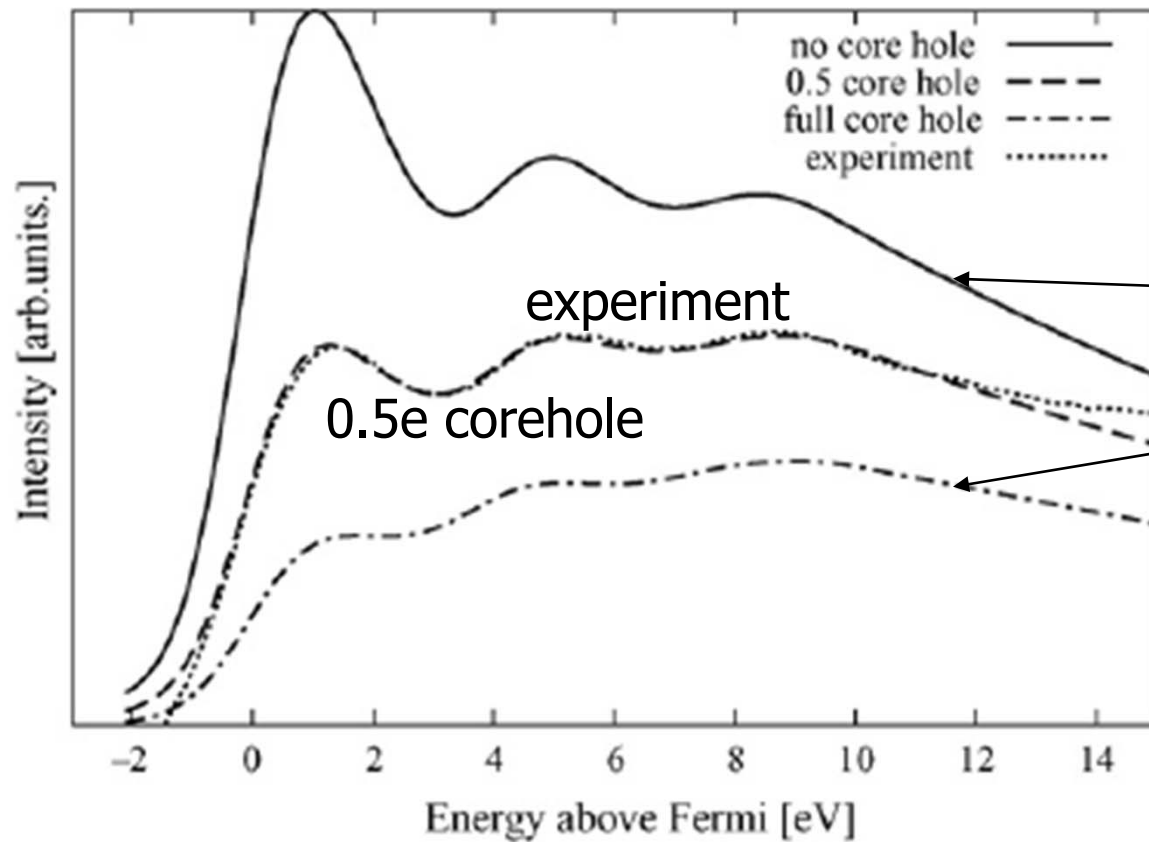




Partial core hole screening in the Cu L₃ edge



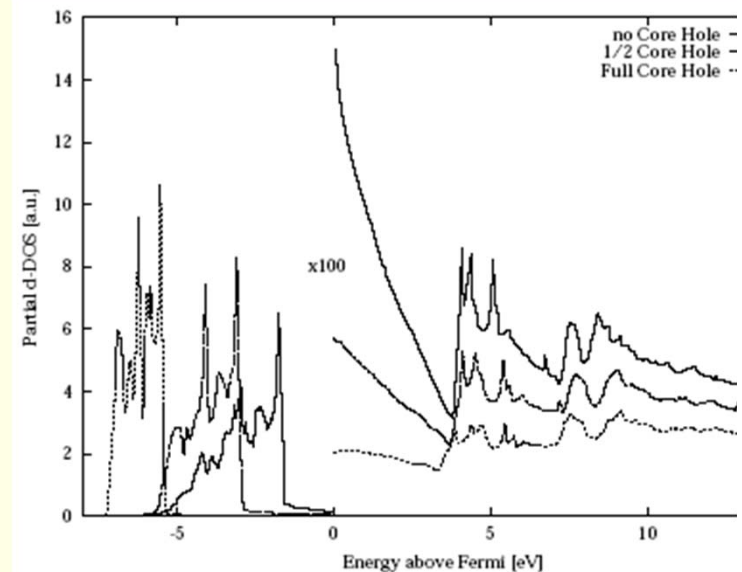
- J.Luitz et al., Eur. Phys. J. B 21, 363{367 (2001)



without
corehole

with
corehole

Unexpected effect of core-hole !

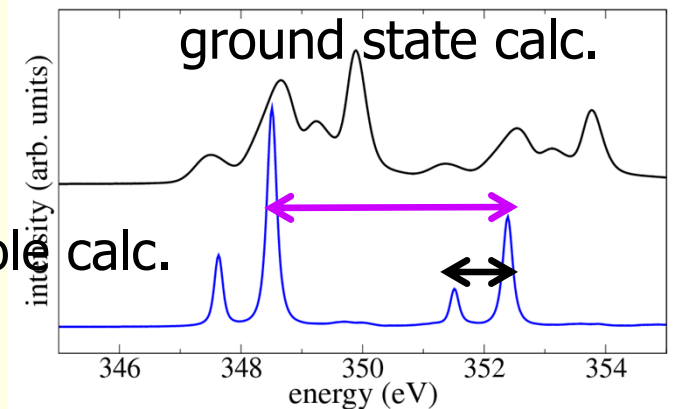
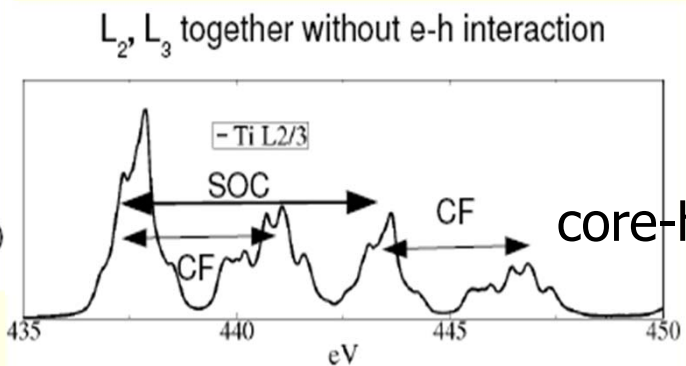
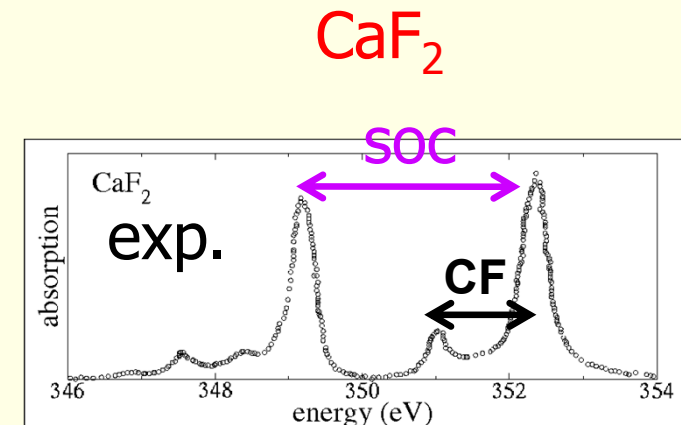
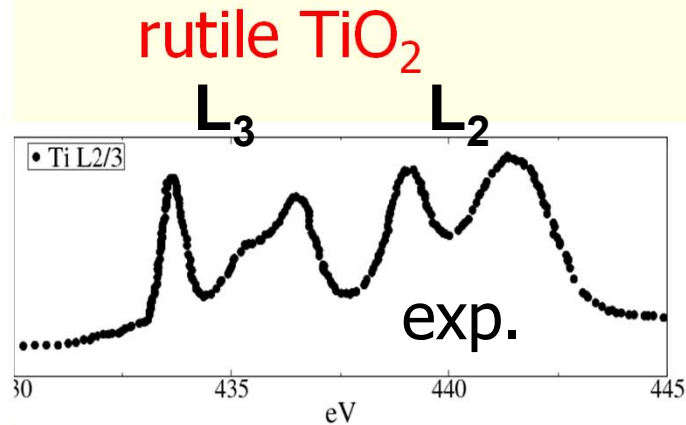
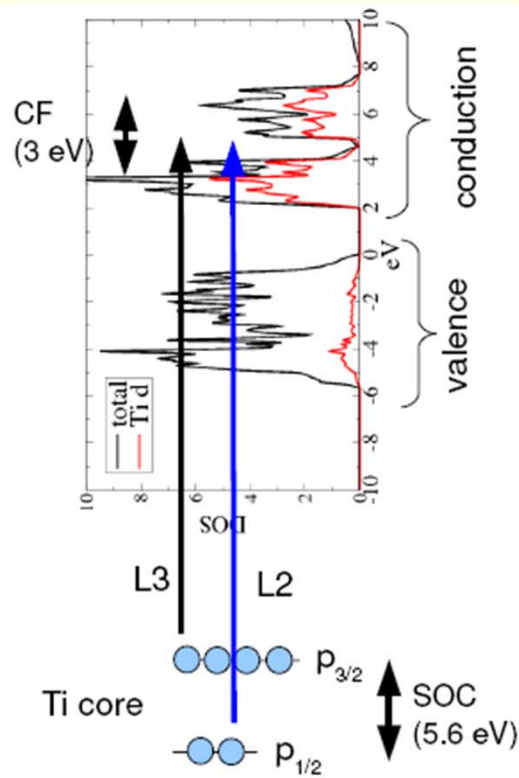




$L_{2,3}$ spectra: failure of the single particle approach



- In particular early 3d TM-compounds show a
 - non-standard L_2/L_3 branching ratio (1:2)
 - sometimes a completely different lineshape (TiO_2)
 - „wrong“ SOC or CF splittings





Beyond IPA



- *ab-initio* configuration interaction (solid is approximated by finite cluster)

Y. Kumagai, H. Ikeno, and I. Tanaka, J. Phys.: Condens. Matter 21, 104209 (2009).

H. Ikeno, F. M. F. de Groot, S. E., and I. Tanaka, J. Phys.: Condens. Matter 21, 104208 (2009).

H. Ikeno and I. Tanaka, Phys. Rev. B 77, 075127 (2008).

- linear response in time dependent DFT (TDDFT)

J. Schwitalla and H. Ebert, Phys. Rev. Lett. 80, 4586 (1998).

A. L. Ankudinov, A. I. Nesvizhskii, and J. J. Rehr, Phys. Rev. B 67, 115120 (2003).

- Bethe- Salpeter equation (BSE)

E. L. Shirley, J. Electron Spectrosc. Relat. Phenom. 144-147, 1187 (2005).

E. L. Shirley, Phys. Rev. Lett. 80, 794 (1998).

J. A. Soininen and E. L. Shirley, Phys. Rev. B 64, 165112 (2001).

W. Olovsson, I. Tanaka, T. Mizoguchi, P. Puschnig, and C. Ambrosch-Draxl, Phys. Rev. B 79, 041102 (2009).

R. Laskowski, P. Blaha, Phys. Rev. B, 81, 075418 (2010)

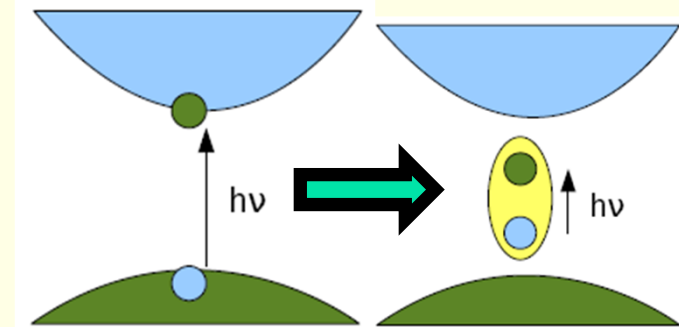


fully relativistic electron-hole interaction (BSE)



- *Bethe-Salpeter-equation: $L(12;1'2')$*
- *solving a 2-particle ($e - h$) equation of large dimension ($N_v N_c N_k \sim 100000$)*

$$\sum_{v'c'k'} (H_{v'c'k',vck}^{eh}) A_{v'c'k'}^\lambda = E^\lambda A_{vck}^\lambda$$



$$H^{eh} = H^{diag} + H^{dir} + 2H^x$$

$$H^{diag} = (E_{v,k} - E_{c,k}) \delta_{cc'} \delta_{vv'} \delta_{kk'}$$

eigenvalue difference between hole (v) and electron(c) state

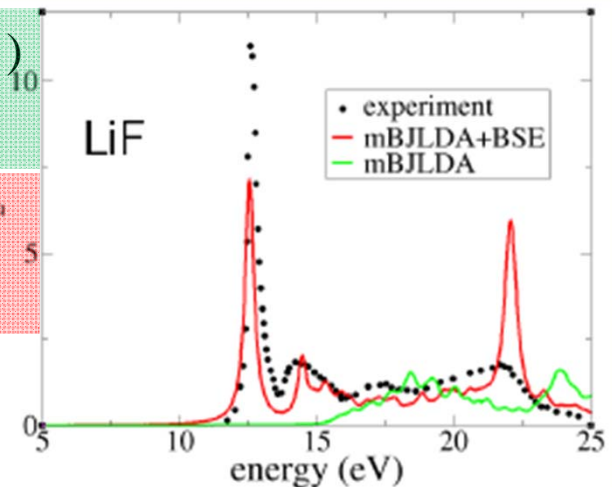
$$H_{vckv'c'k'}^{dir} = - \int d^3r d^3r' \Psi_{vk}(r) \Psi_{ck}^*(r') W(r,r') \Psi_{v'k'}^*(r) \Psi_{c'k'}(r')$$

attractive screened static Coulomb interaction W ; $W \sim \epsilon^{-1}$

$$H_{vckv'c'k'}^x = \int d^3r d^3r' \Psi_{vk}(r) \Psi_{ck}^*(r) \bar{v}(r,r') \Psi_{v'k'}^*(r') \Psi_{c'k'}(r')$$

e-h exchange with bare Coulomb potential v

Excitons in LiF





Ca-L₂₃ edge in CaF₂



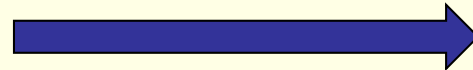
■ “ground-state” DOS



■

■

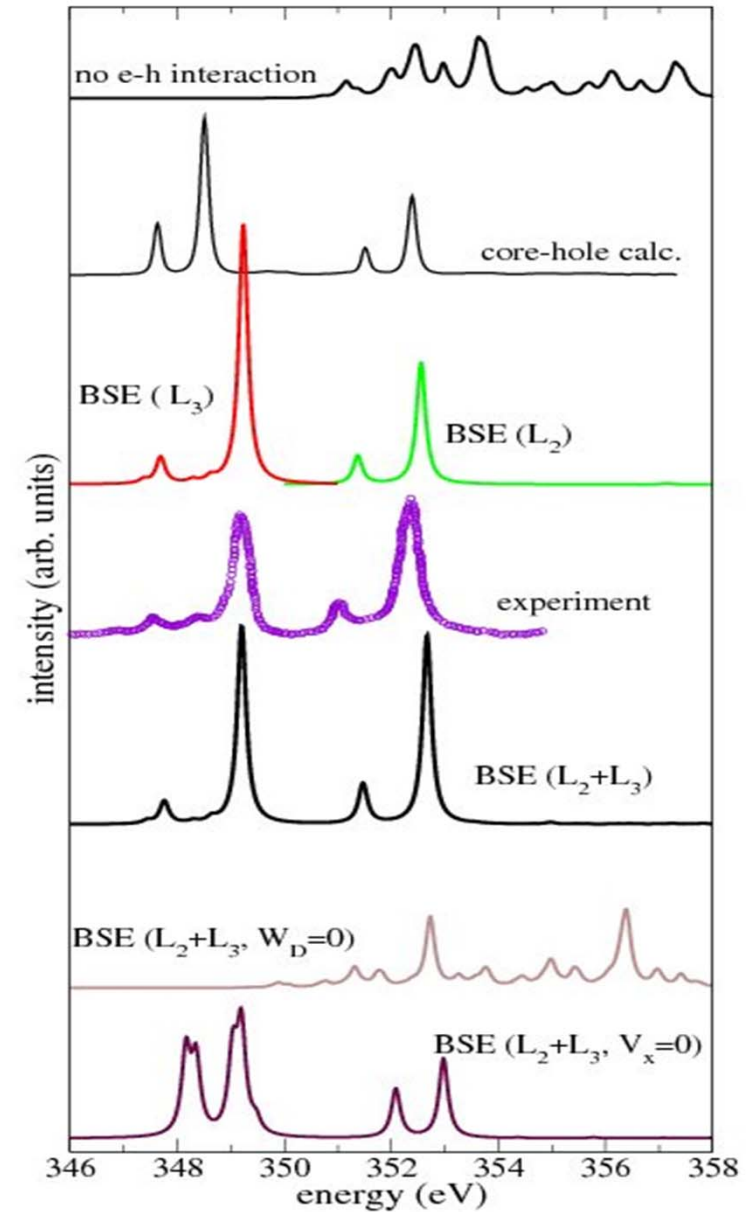
■ experiment



■

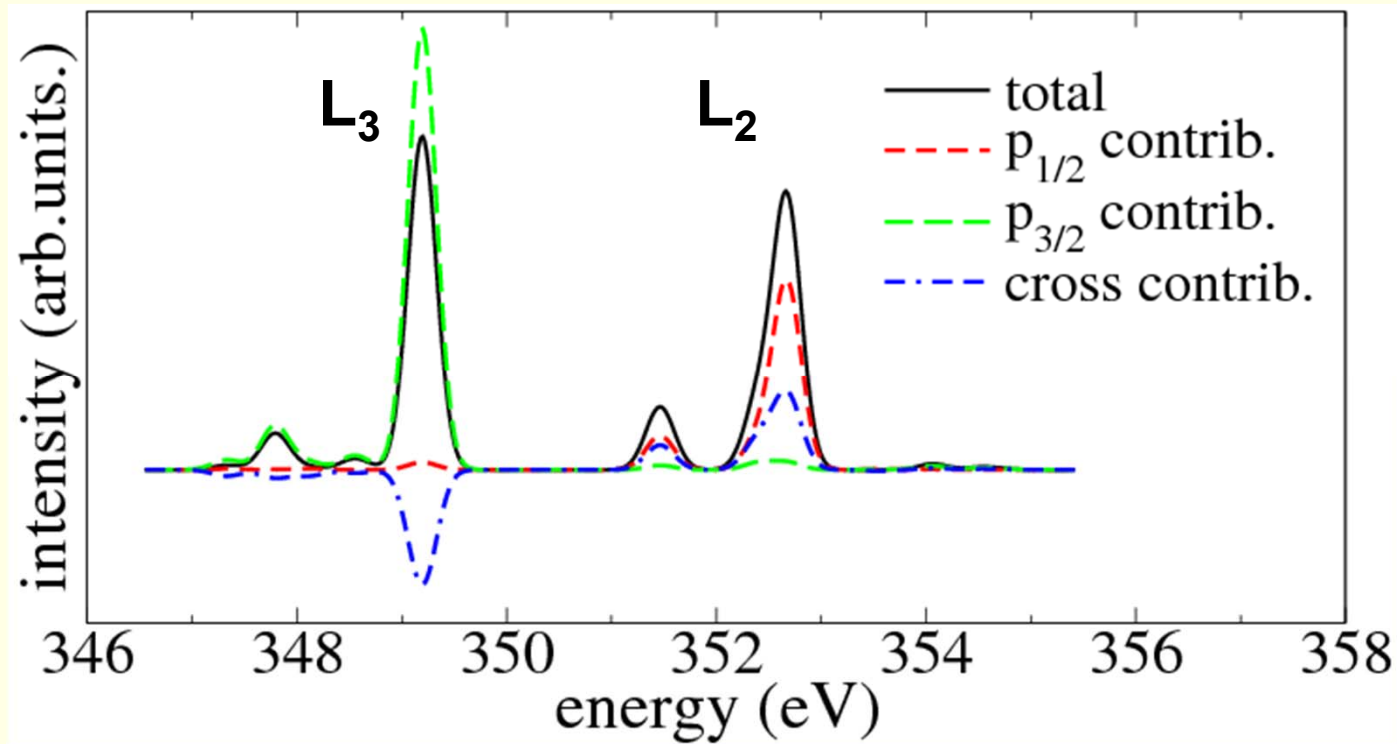
■

■





$L_{2,3}$ edge for Ca in CaF_2

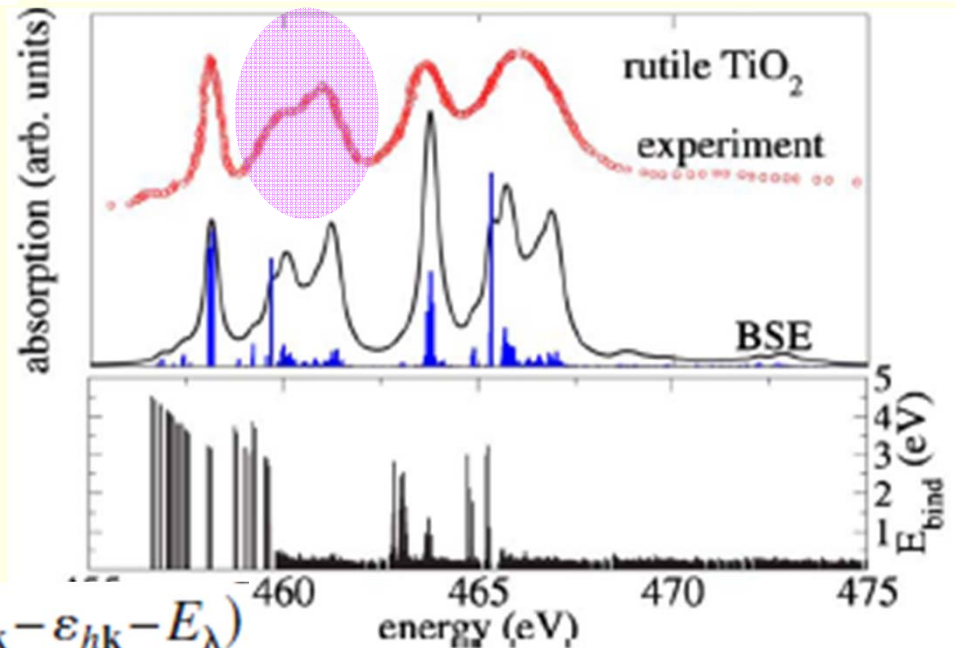
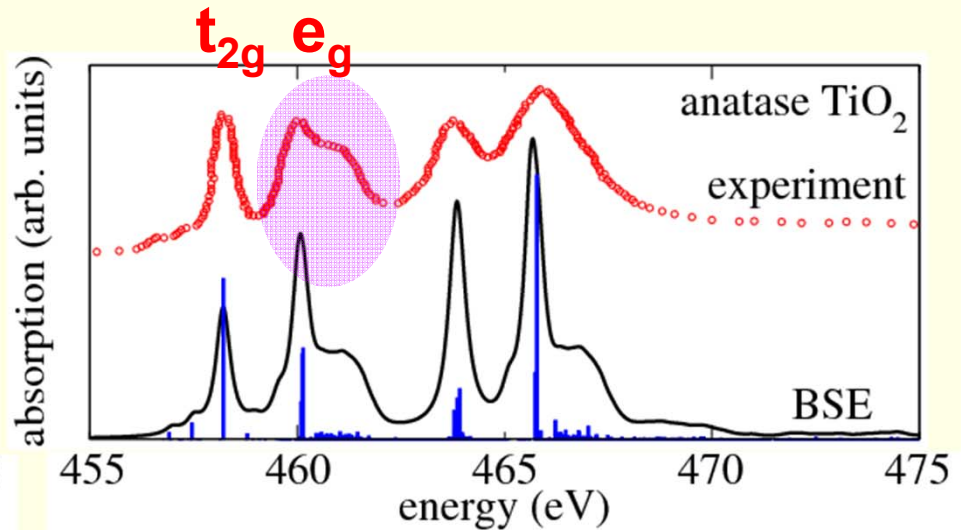
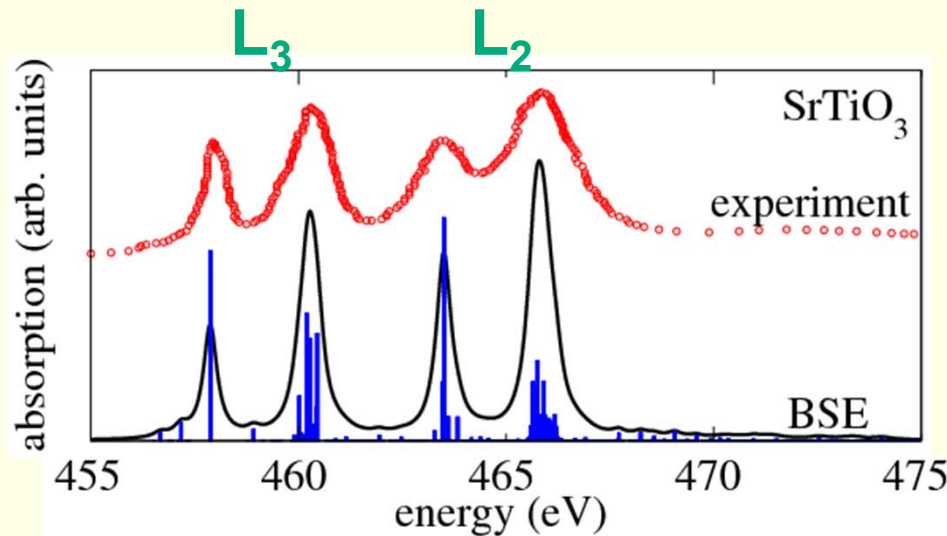


Decomposition of ϵ_2 into the excitation from $p_{1/2}$ and $p_{3/2}$ states
cross terms suppress the L_3 branch and enhance L_2

$$\epsilon_2^{xx}(\omega) = \frac{8\pi^2}{\Omega} \sum_{\lambda} \left| \sum_{hek} A_{hek}^{\lambda} \frac{\langle hk | -i\nabla_x | ek \rangle}{\epsilon_{ek} - \epsilon_{hk}} \right|^2 \times \delta(E^{\lambda} - \omega)$$



Ti L_{2,3} in rutile-TiO₂, anatase-TiO₂, SrTiO₃



- The experimental Ti L_{2,3} edges are rather well reproduced.
- intensity ratio L₃/L₂ (not 2:1)
- „t_{2g}/e_g“ ratio (not 3:2)
- left/right shoulder in L₃-, „e_g“ peak of rutile/anatase

$$E_{bind}^{\lambda} = \sum_{k,e,h} A_{k,h,e}^2 (\epsilon_{ek} - \epsilon_{hk} - E_{\lambda})$$



Ti $L_{2,3}$ in SrTiO_3 , decomposition of ϵ_2

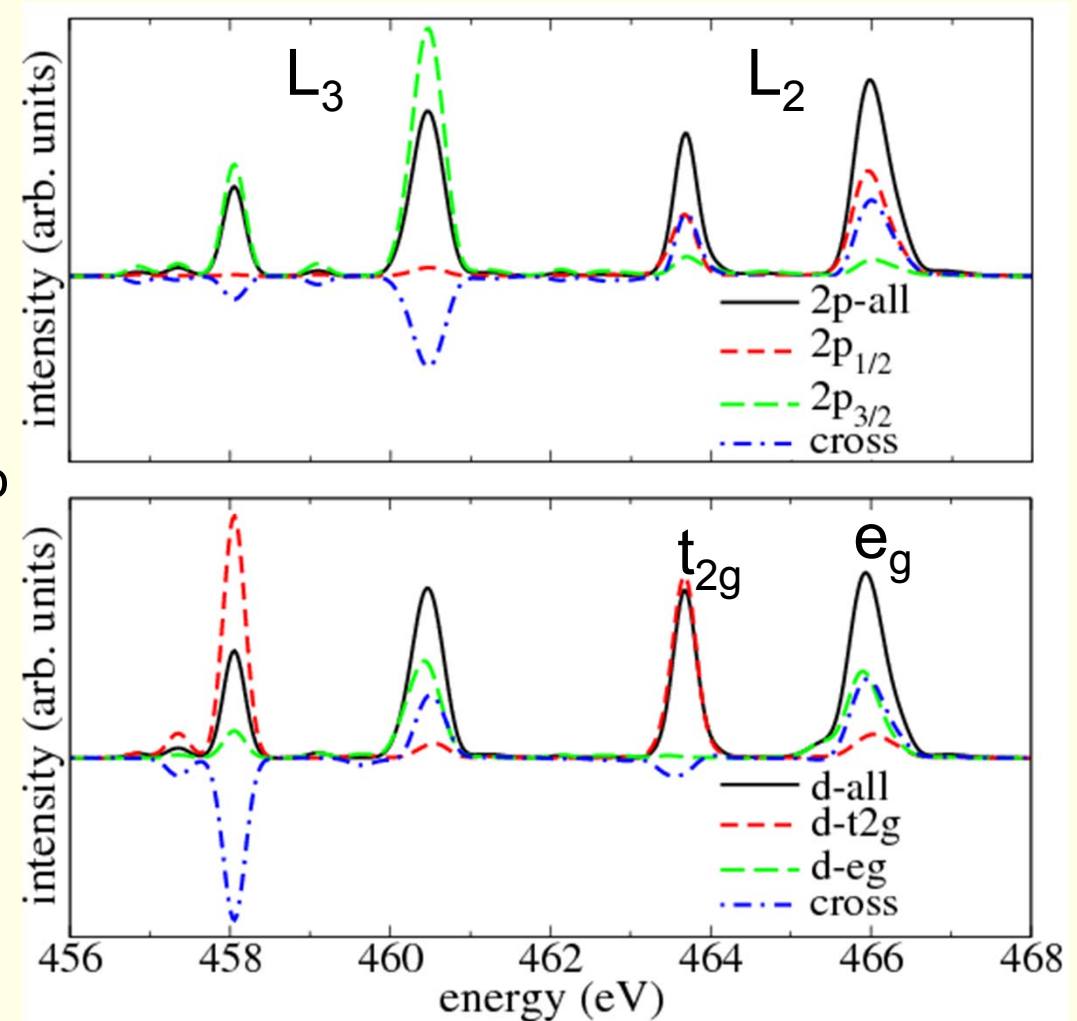
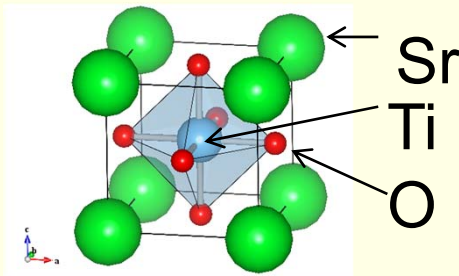


Decomposition into excitations from $p_{1/2}$ and $p_{3/2}$

cross term suppresses the L_3 branch and enhances the L_2

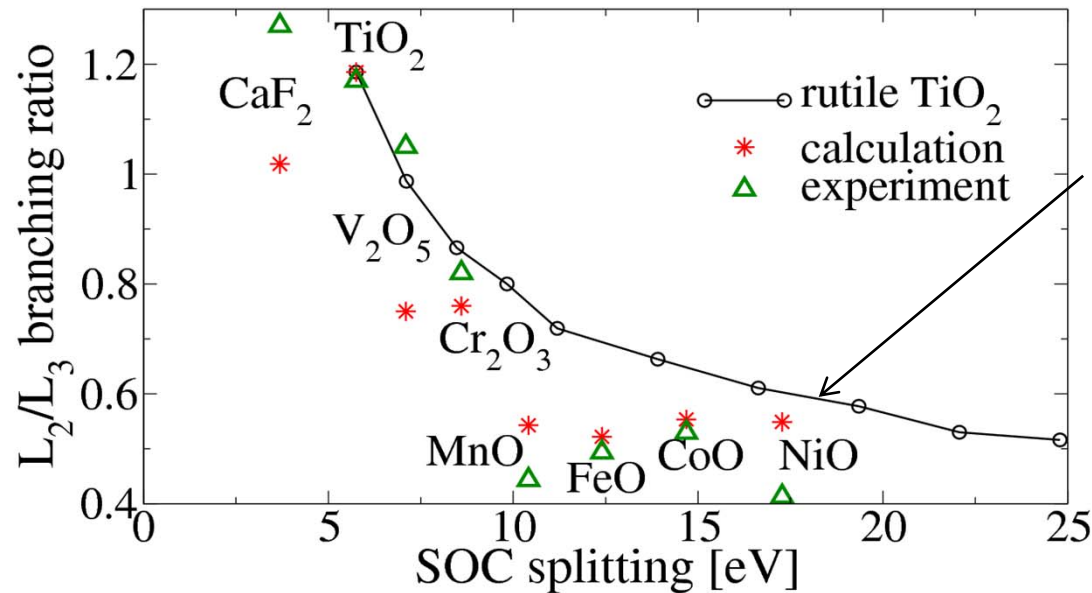
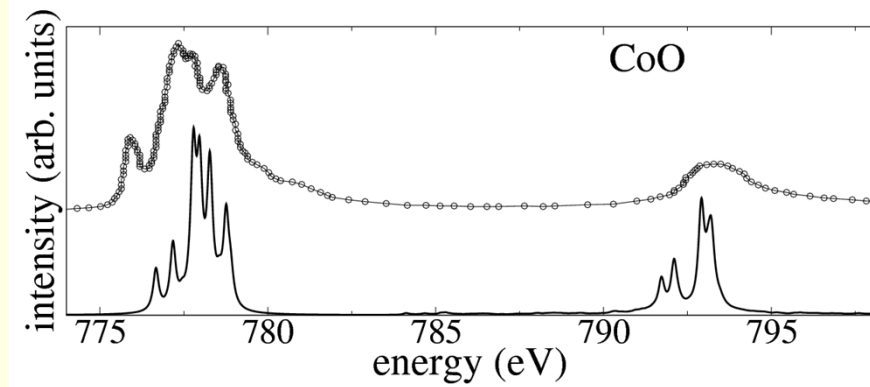
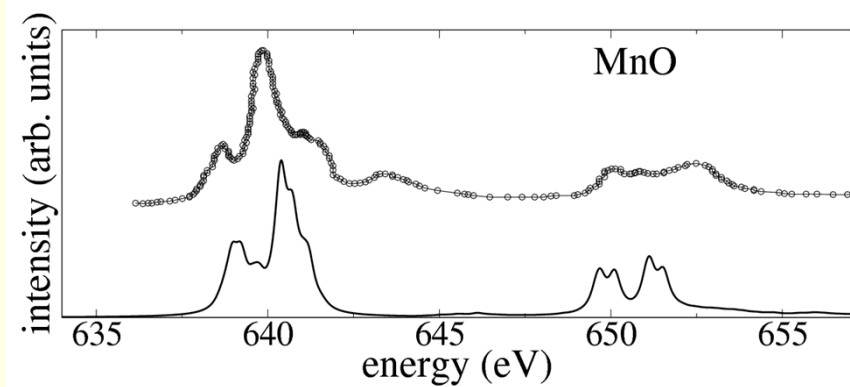
Decomposition into excitation into the e_g and t_{2g} bands

the first peak of L_2 or L_3 is related to excitations into t_{2g} , the second peak is associated with e_g
the cross term strongly modifies the ratio between these peaks





3d metal $L_{2,3}$ branching ratio

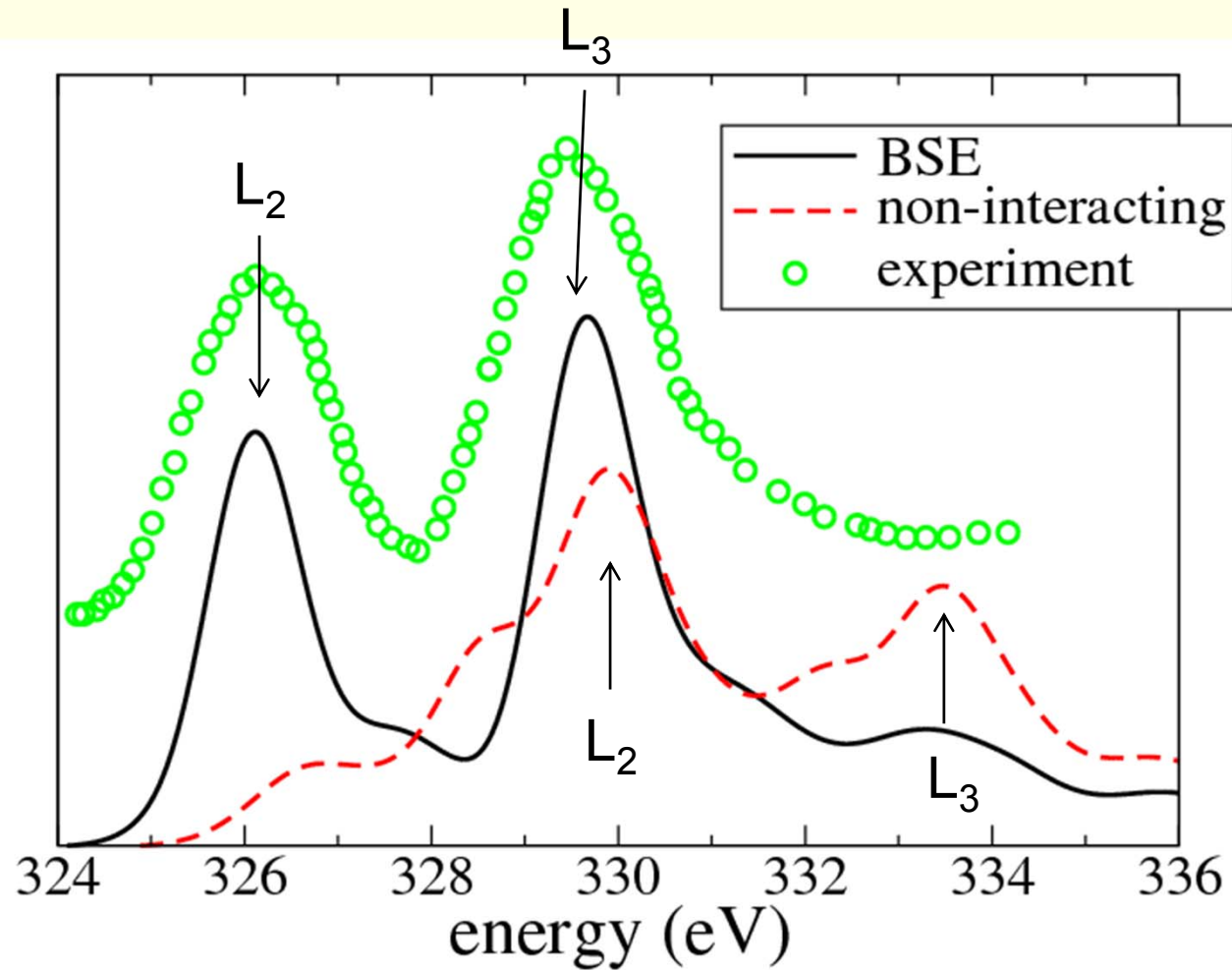


Ti $L_{2,3}$ branching ratio in rutile-TiO₂
(SOC splitting is set to values from 5eV to 25eV)

The size of **spin orbit splitting** is a main factor determining the **branching ratio** (deviating from 0.5)



fcc Ca, $L_{2,3}$ edge





Acknowledgement



“Practical aspects of running the WIEN2k code for electron spectroscopy”,
C.Hebert, Micron 38 (2007) 12–28

Acknowledgement:

TELNES: P.Schattschneider, M.Nelhiebel, C.Hebert (TU Vienna)
K.Jorissen (Univ. Washington)

BSE: R.Laskowski (TU Vienna)
C.Ambrosch-Draxl



WIEN2k: K.Schwarz, J. Luitz

Thank you for your attention !